

Examination the antibacterial activity for the TiO₂ nanoparticles which is prepared by hydrothermal method and mixed with plant and animal extracts

Falah H. Ali *

Muslim A. Abid

Wisam J. Aziz

¹Department of Physics, College of Science, Mustansiriyah University, Baghdad- Iraq

^{*}Corresponding Author E-mail: falahic3@gmail.com

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ABSTRACT

Initial production of the titanium dioxide nanoparticles TiO₂ used the hydrothermal process. The onion peels and chitosan peel extracts, which are made using the Bio-synthesize approach. By using (UV-Vis) properties, the structural characteristics of TiO₂ and TiO₂ with extract from onion peels and chitosan peels were studied, and find the absorption edge of TiO_2 NPs, TiO_2 + onion peels and TiO_2 + chitosan peels is around 236 nm, 244 nm and 237.5 nm respectively. The pattern of TiO_2 for a number of peaks suggested a tetragonal rutile phase, according to X-ray diffraction (XRD) research. Energy dispersive Xray, (EDX) was done for TiO_2 NPs, TiO_2 + onion peels and TiO_2 + chitosan peels. Field emission scanning electron microscopy (FE-SEM) shows that the diameters ranging of TiO_2 NPs, TiO_2 + onion peels and TiO_2 + chitosan peels from (43.32 nm to 47.77 nm), (34.11 nm to 62.77 nm) and (32.22 nm to 35.11 nm) respectively. Studying was done on the effect of TiO₂ & TiO₂ with extracts nanoparticles on bacterial growth, a variation was observed in the effect of the TiO₂ & TiO₂ with extracts on the growth of bacteria.

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المستخلصات النباتية والحيوانية				
وسام جعفر عزيز	مسلم عز الدين عبد	فلاح حسن علي		
الكلمات المفتاحية:	م الفيرياء كليه العلوم - الجامعة المستنصرية- بعداد- العراق	الندلك		
مستخلص قشور الشيتوزان، الطريقة الحرارية المائية، مستخلص قشور البصل، ثاني أكسيد التيتانيوم TiO ₂ .	النانوية TiO2 أستخدم العملية الحرارية المائية. يتم تصنيع متخدام نهج التخليق الحيوي. وباستخدام خصائص (-UV 5 وTiO2 مع مستخلص قشور البصل وقشور الشيتوزان، TiO2 + قشور البصل وTiO2 + قشور الشيتوزان تبلغ نانومتر على التوالي. يشير نمط TiO2 لعدد من القمم إلى	اول انتاج لجسيمات ثاني أكسيد التيتانيوم قشور البصل ومستخلصات قشر الشيتوزان باس (Vis) تمت دراسة الخصائص التركيبية لـ FiO2 ووجد أن حافة الامتصاص لـ TiO2 NPs و2 حوالي 236 نانومتر. و 244 نانومتر و 237.5		

فحص النشاط المضاد للبكتيريا لجزيئات TiO2 النانوية المحضرة بالطريقة الحرارية المائية والمخلوطة مع المستخلصات النباتية والحيوانية مرحلة الروتيل رباعي الزوايا، وفقًا لأبحاث حيود الأشعة السينية (XRD). تم إجراء الأشعة السينية المشتنة للطاقة (EDX) لـ TiO2 NPs وTiO2 ح تشور البصل وTiO2 + قشور الشيتوزان. يُظهر المجهر الإلكتروني الماسح بالانبعاث الميداني (FE-SEM) أن أقطار NPs 102 وTiO2 + قشور البصل وTiO2 + قشور الشيتوزان تتراوح من (43.32 نانومتر إلى 47.77 نانومتر)، (34.11 نانومتر إلى 62.77 نانومتر) و (32.22 نانومتر إلى 35.11 نانومتر) نانومتر) على التوالي. أجريت دراسة تأثير TiO2 وTiO2 وTiO2 بالمستخلصات النانوية على نمو البكتيريا، ولوحظ اختلاف في تأثير TiO2 وTiO2 وTiO2 وTiO2 وTiO2 بالمستخلصات على نمو البكتيريا.

1. INTRODUCTION

Nanocomposites (NCs) have attained a considerable attention by the academia and businesses since they can be used to make new, improved materials that can be used in a wide range of fields, such as electrical engineering [1]. Additives are components that can be added to polymers in order to impart specific properties and improve fundamental properties[2]. Although the majority of commercial polymer mixtures are produced by melting and mixing melts in twin-screw extruders, the majority of experimental studies have been conducted by casting from solutions. The interactions between polymer chains determine the miscibility of two polymers[3]. Polyvinyl alcohol (PVA) is a manufactured thermoplastic polymer that is white and has no smell. Its chemical formula is $(C_2H_4O)n$. [4]. It is a material that is safe and works with living things. It works well as an insulator, an optical material, and a charge storage material. This polymer is used in a wide range of businesses, including adhesives, coatings, electronics, construction, textiles, paper, and biomedical. Most of the time, vinyl acetate monomer is used to make PVA for sale. In alkaline conditions, hydroxyl groups replace the ester groups in polyvinyl acetate (PVA) [5]. The thermoplastic polymer of poly (ethylene glycol) (PEG) type has a flexible bond structure which its chemical formula is H–(O–CH₂–CH₂)_n–OH [6].

Nickel (III) Oxide (Ni_2O_3) is a semiconducting p-type material with an energy band gap of almost 2.2-3.4eV [7, 8]. Sol-gel, solvent-thermal processes, and hydrothermal

synthesis are only some of the methods that can be used to create this potential metal oxide. One such technique that has garnered a lot of interest thanks to its applications in gas sensors, catalysis, ferrofluids, and magnetic storage devices is low-temperature solution combustion synthesis. The structure of Ni₂O₃ is also orthorhombic, which makes it interesting to study [9]. Blending is the physical combining of two or more polymers to create a new material whose properties depend on the polymer composition [10]. Blending, which frequently results in polymers with superior behavior than their component polymers, is one of the most effective methods to improve various physical properties [11]. His work provides an effective way to improve the morphology, optical and AC electrical properties of (PVA-PEG) and the nanocomposites by loading different ratios of Ni₂O₃ NPs.

2. Experimental part:

2.1. Materials

PVA [partially hydrolysis] of 160,000 g/mol was supplied by Alpha Chemika, India, and PEG [degree of hydrolysis= 99%] of 20,000 g/mol was provided by Central Drug House, Ltd, Company, India, are applied as granular form. Ni₂O₃ (Sigma Aldrich) black powder with a particle size of \leq 40 nm and purity of 99.8%, (74.69 g/mol.) was insoluble in water was used.

2.2. Purification of NC films.

Briefly, 0.9 g of PVA in 50 ml distilled water (DW) was dissolved for 1 hr at RT, then this was continued for another period under 75-80°C using magnetic stirrer. After cooling the

solution to 40 °C, 0.1g of PEG was added to synthesis the polymer blend. The resulting solution was poured into a plastic Petri dish and kept under air at RT for 240 hours to drying. The same method was followed in the preparation of PVA-PEG/ Ni₂O₃ NC films. The method presented in Table 1. The thicknesses of the prepared films were about ($120 \pm 4 \mu m$), as measured by Digital Vernier Caliper.

Table 1:	Selected	values	of PVAs	, PEGs	and	NC
films						

Sample	PVA (g)	PEG (g)
PVA-PEG	0.9	0.1
3 wt.% Ni ₂ O ₃	0.873	0.097
4.5wt.% Ni ₂ O ₃	0.8595	0.0955
6 wt.% Ni ₂ O ₃	0.846	0.094
7.5 wt.% Ni ₂ O ₃	0.8325	0.0925
9 wt.% Ni ₂ O ₃	0.819	0.091

2. 3 Descriptions

Fourier transform infrared spectroscopy (Bruker company, type vertex -70 spectrometer, German origin) was used to show the chemical components of the synthesized samples in the range of 4000–500 cm⁻¹ at RT. Field Emission Scanning Electron Microscope (FESEM) was used to examine the surface. Nanocomposite formation was analyzed on a UV-visible spectrophotometer (Shimadzu UV-1650 PC, Phillips, Japanese company) at 200 -1100 nm. The dielectric characteristic were studied at f=10² to 5 ×10⁶ Hz by LCR meter (HIOKI 3532-50 LCR HI TESTER).

3. Results and discussion

3.1 Fourier transform infrared (FT-IR)

FTIR spectra of PVA-PEG blend and its NC films based on Ni₂O₃ NPs shown in the Figure1. It exhibits eight major absorption peaks. The broad peak at 3278.90 cm⁻¹ that assigned to the stretching vibration of the alcohol group (OH) in the polymer matrix chain [8]. The two intense peaks at 2908.33, and 2360.00 cm^{-1} corresponds to the Methyl C-H₃ asymmetric and strong O=C=O stretching vibration [9]. Additionally, the peak at 1716.51 cm^{-1} corresponds to the C=O stretching band, which is actually attributed to carboxylic acid, aliphatic ketone, aldehyde, or quinine groups [10]. The peak at 1340.55 cm^{-1} is attributed to the deformation stretching vibration of the strong N-O . The peak at 1239.16 cm^{-1} is deformation stretching attributed to the vibration of the C-N link [8]. Moreover, the peaks observed at 1085.02, and 830.52 cm^{-1} could be attributed to the twisting vibration of strong C-O-C and medium C=C bending vibrations [11]. As was mentioned before, the intensity of some of these bands decreased and others increased to refer to strong interficial physical interaction between the host and the embbeded nanopartical with small shifts toward higher wavenumbers, which is a sign that the hydrogen bonds were formed by the physical interaction between the functional groups in the polymer blend and Ni₂O₃ NPs. Strong incorporations with Ni⁺² ions are another potential cause of intensity alterations and broadening of the functional groups in the polymeric matrix's backbon [12]. The peaks for the PVA-PEG /Ni₂O₃ (Figure1,b-f) are alike to the spectrum of PVA-PEG, showing this blend polymer make up a big part of the NCs. This can verify that the synthesis was effective of the PVA-PEG /Ni₂O₃ NCs.



Fig. 1: FT-IR of blended polymer PVA-PEG (a) with 3wt.% (b) , 4.5wt.% (c) , 6wt.% (d) , 7.5wt.% (e), and 9wt.% (f) of Ni_2O_3 NPs

3.2 Field emission scanning electron microscopy

FESEM analysis was carried out on blended polymer and its NC films based on (3, 4.5, 6, 7.5 and 9) wt.% of Ni_2O_3 NPs to show the surface morphologies of prepared films as seen in Figure 2. The images were taken at a magnification of 20.00 kx. The films show many aggregates or chunks randomly distributed of NPs on the films surface. The results show an increase in the number of aggregations on the surface in accordance with increasing the amount of Ni_2O_3 .











Fig. 2: FESEM of blended polymer (a) pure PVA-PEG, (b) 3wt.%, (c) 4.5wt.%, (d) 6wt.% (e) 7.5wt.%, and (f) 9wt.% of Ni_2O_3 NPs

3.3 Optical investigation

The transmittance spectra of blended polymer and its NCs with (3, 4.5, 6, 7.5, and 9)wt.% Ni_2O_3 films are shown in Figure (3). Above 300 nm, the transmittance curves of all samples show a tendency towards saturation, and the highest rated average transmittance of the blended polymer is ~ 88% in the Vis and NIR areas of spectrum, but it decreases almost gradually with increasing the weight reaching to 66% at 9 wt.% Ni₂O₃. This feature was ascribed to the film surface morphology and the Ni₂O₃ contribution absorption. exhibited absorbance peaks at ~ 280 nm, which could appear through the charge transfer from the ligand to the metal. Also, the decrease in the transmittance of the prepared films in UV region comes from increasing the nanoparticle concentrations that makes it suitable for different applications, as packaging for storage drugs regardless of cost or for solar cell applications.





The relationship in equation 1 was used to derive the absorption coefficient [12]

$$\alpha = 2.303 \frac{A}{t} \tag{1}$$

Where A is the absorbance and t are the thickness of the film. From Fig. 4, it is noted the shift in the absorption edge towards the lower energy with an increase in the amount of the additive. This finding show that as the percentage of the additive goes up, the optical energy gap (E_g^{opt}) goes down.





Indirect E_g^{opt} is related to the photon energy (*hv*) and the α , as shown in equation 2 [13]. (α hv= B (hv - E_g^{opt} \pm Eph)^{r} (2) Where B is the parameter for band tailing, and r defines the type of optical transition for the materials under investigation (r=2 for allowed indirect transitions). The optical band gap energy can be obtained from the plot of $(\alpha hv)^{1/r}$ versus photon energy (hv) shown in Figure 5 are named as illustrated in Table (2):



Fig. 5: Variation of E_g^{opt} of blended polymer and its NCs.

Sample	Allowed (eV)
PVA-PEG	4.385
3wt.% Ni ₂ O ₃	4.28
4.5wt.% Ni ₂ O ₃	4.10
6 wt.% Ni ₂ O ₃	3.92
7.5 wt.% Ni ₂ O ₃	3.57 and (2.31)
9 wt.% Ni ₂ O ₃	3.39 and (2.22)

Table 2: E_g^{opt} indirect transition of blended polymer and its NCs.

The indirect band gaps obtained with procedure PVA-PEG/ 3-9wt.%Ni₂O₃ NC films, may be calculated by projecting the linear portion of the curve to the hv axis. It can be observed that allowed indirect E_g^{opt} decreased with additives of Ni₂O₃ NPs. Moreover, the samples with a higher concentration of 7.5 and 9 wt. % of Ni₂O₃ NPs showed two band gaps compared to other samples. The increase in oxygen vacancies (V0) which occurred just below the conduction band may cause reduction in optical energy gap [8]. These positively charged vacancies may capture some

free electrons and act as donor centers leading to the reduction in optical energy gap. From another point of view, the formation of imperfections and disarrays into the materials, which is close to the conduction band, may explain why the E_g^{opt} decreased. As a result, it can soak up low-energy photons. These matches the lookup values [14,15]. It's noticed that in the optimum incorporating amount (Ni₂O₃ NPs = 7.5 and 9 wt.%), there are two band gaps that appear; the first is correlated with the pristine composition elements, and the second may be correlated to the high wt.% of the nanoparticles, which generated the new secondary energy levels. The index of refractive (n), polarizability (P) and extinction coefficient (K_0) of PVA-PEG/3-9wt.% Ni₂O₃ NC films were considered from the equations 3,4, and 5 [16, 17].

$$n = \frac{1+R}{1-R} + \left[\frac{4R}{1-R^2} - k_{\circ}^2\right]^{1/2}$$
(3)
$$P = \frac{3}{4\pi} \left(\frac{n^2 - 1}{n^2 + 1}\right)$$
(4)

$$K_0 = \frac{\alpha \lambda}{4\pi} \tag{5}$$

When developing a variety of electronic devices, such as those for optoelectronic, photonic, and cable applications, the evaluation of refractive index (n) is thought to be a crucial factor [16]. Figure 6 elucidates the dependence of (n) on different ratios of Ni₂O₃ NPs in blended polymer PVA-PEG. From the photodarkening phenomenon, the (n) was found to be increased with the increase in the amount of Ni₂O₃ NPs, and the structural change criteria is what causes the fluctuation in the value of (n). The values of refractive index showed small increases from 2.45 to 2.68 with the increasing concentration of nanoparticles from 3 to 9 wt.%.



Fig. 6: Index of refractive for blended polymer PVA-PEG and its NCs.

According to the literature [17], the larger the P, the greater the n, and non-polarizing materials don't change the speed of light (meaning there is no change in the internal structure after addition), so n = 1. Decreasing in the E_g^{opt} with the increase the ratio of the additive Ni₂O₃ NPs means that the electrons can move to other levels and thus an increase in the polarizability of the material as shown in Fig. 7. Due to the dipoles created being unable to maintain the high frequency, polarization can be seen to diminish at lower wavelengths as the incident photon's energy increases (high The polarizability values energies). were increased upon loading of Ni₂O₃ NPs.



Fig. 7: Polarizability of blended polymer PVA-PEG and its NCs.

According to our analysis of Figure 8, the extinction coefficient results of the NC films are much higher than those of the blended polymer in all regions. This behavior attributed to increase of the absorption of the incident light [18]





The optical conductivity (σ_{op}) depends directly on the refractive index (n) and absorption coefficient (α) by via the relative 6

[19].

$$\sigma_{\rm op} = nc\alpha/4\pi \tag{6}$$

where c is the velocity of light. Figure 9 elucidates the dependence of optical conductivity on different ratios of Ni₂O₃ NPs in blended polymer. It's noticed that the highest optical conductivity values appears in the UV region particularly in optimum incorporating amount of Ni₂O₃ NPs (9 and 7.5 wt.%). This tendency results from the band structure's localized stages becoming denser, which raises the absorption coefficient and, in turn, the optical conductivity. σ_{op} spectra confirm the transmittance behavior within the Vis and NIR regions.



Fig. 9: Optical conductivity of blended polymer PVA-PEG and its NCs.

3.4 The AC electrical properties

The dielectric constant of blended polymer and the NC films at 100Hz are shown in Figure 10. The dielectric constant was calculated using equation 7 [20]:

$$\dot{\epsilon} = Cp \quad d/ \quad \epsilon_o \quad A$$
(7)

where, Cp is capacitance of matter, d is the width in cm, and A is the area in cm^2 . As the number of Ni2O3 NPs goes up, as shown in the figure ,so the dielectric constant of NCs increases. This behavior can be explained by the polarization of the interfaces between the NCs when an alternating electric field is applied and the rise in charge carriers [21]. Figure 11 shows that as the frequency goes up, the space charge as a percentage of the total polarization polarization goes down. This makes the dielectric constant goes down for all PVA-PEG/Ni2O3 NC films, so space charge polarization contributes most to polarization at lower frequencies.



Fig.10: Effect of Ni₂O₃ on dielectric constant for PVA-PEG/Ni₂O₃ at 100Hz.



Fig.11: Variation of dielectric constant for $PVA-PEG/Ni_2O_3$ with frequency at RT.

The sample's dielectric loss is the amount of electrical energy lost due to the transformation of the applied field into thermal energy. The dielectric loss of NCs can be calculated using equation 8 [22]:

(8)

D is dispersion factor. Dielectric loss as a function of electric field frequency of Ni_2O_3 NPs in the PVA-PEG blends at RT shown in Figure 12. The figure show that the dielectric loss of NCs decreases with increasing of electric field frequency for all samples. The decrease in the space charge polarization contribution is responsible for this behavior. In addition, the maximum dielectric loss for 9 wt.% Ni_2O_3 at low frequency (10^2 Hz) is 0.0692. The dielectric loss of NCs based on Ni_2O_3 increases with the increasing the number of NPs as shown in Figure 13. This was associated with a rise in the total number of charge carriers. Similar behavior was reported in [23].



Fig. 12: Variation of dielectric loss for PVA-PEG/Ni₂O₃ with frequency at RT



Fig. 13: Effect of Ni_2O_3 on dielectric loss for PVA-PEG blends at 100Hz.

The A.C electrical conductivity of NCs was calculated using equation 9 [24]:

$$\sigma_{A,C}=2\pi f$$
 $\acute{\epsilon}$ D ϵ_{o} (9)

Figure 14 shows how the AC electrical conductivity of Ni₂O₃ NPs in PVA-PEG mixes depends on the frequency of the electric field. As the frequency of the electric field increases, the AC conductivity increases dramatically for all samples. The polarization of space charges at low frequencies and the hopping motion of charge carriers are responsible for this behavior [23]. Also the conductivity increases with the increasing wt.% of Ni₂O₃ in the PVA-PEG blends at 10^{2} Hz as shown in Figure 15. Incorporating additive NPs into the composition lowers the resistance of NCs and raises their A.C electrical conductivity by increasing the number of charge carriers. Similar behavior was reported in [23].



Fig. 14: Variation of A.C electrical conductivity for PVA-PEG/Ni₂O₃ with frequency at RT



Fig.15. Effect of Ni_2O_3 on A.C electrical conductivity for (PVA-PEG) blend at 100Hz.

4. Conclusions

FTIR spectra confirms the presence major functional groups belonging to the polymer systems. The SEM images of the films show or chunks many aggregates randomly distributed of NPs an increase in the number of aggregations on the surface in accordance with increasing the amount of Ni₂O₃. Above 300 nm, the transmittance curves of all samples show a tendency towards saturation, and the value for blended polymer film was ~ 88% in the Vis and NIR areas of spectrum, but it decreases almost gradually with increasing the weight reaching ~66% at 9 wt.% Ni₂O₃. The allowed indirect E_g^{opt} decreases with the increase of Ni₂O₃ NPs. Change of electrical parameters such as dielectric constant, dielectric loss, and A.C electrical conductivity for PVA-PEG/Ni₂O₃ as a function of Ni₂O₃ concentrations and applied electric field frequency makes its suitable for application in capacitors, transistor, and electronic circuits.

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